# Monomer and Polymer Solvent Dynamic Control of an Electron-Transfer Cross-Reaction Rate at a Redox **Polymer/Solution Interface**

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Abstract: The electron-transfer cross-reaction rate constant  $k_{12}$  for the thermodynamically disfavored oxidation of the metal complex solute  $[Fe(4,7-dimethylphenanthroline)_3]^{2+}$  by a poly $([Os(bpy)_2(vpy)_2]^{3+})$  redox polymer surface has been measured with a microelectrode-based voltammetric method as a function of the solvent in which the iron complex is dissolved. In the monomeric solvents acetonitrile, acetone, methylene chloride, dimethyl sulfoxide, pyridine, dimethoxyethane, and propylene carbonate, variations in  $k_{12}$  follow a theoretical model incorporating the monomer solvent relaxation time, dielectric constant, and reaction free energy. The osmium metal complexes at the redox polymer/solvent interface are shown to be well solvated without significant constraints on solvent properties.  $k_{12}$  is much smaller when the [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> complex is dissolved in the poly(ether) solvent CH<sub>3</sub>O(CH<sub>2</sub>CH<sub>2</sub>O)<sub>8</sub>CH<sub>3</sub>, Me<sub>2</sub>PEG-400. In the context of solvent dynamics theory, the results indicate that the relaxation time for the polymer chain segment's dipole reorientation that influences the barrier-crossing frequency is ca. 22 ps. This measurement is a new approach to polymer solvent dynamics and is the first example of an outer sphere electron-transfer cross-reaction rate in a polymeric solvent.

The effects of the solvent on electron-transfer reaction rates have occupied considerable recent research attention.<sup>1</sup> The influence of equilibrium aspects of the solvent medium on the barrier free energy for electron transfer was recognized long ago,<sup>2,3</sup> but the role of nonequilibrium aspects of the solvent in barrier-crossing dynamics has only recently been treated. Current theoretical models predict<sup>4</sup> that both dynamic and dielectric solvent properties exert strong influences on electron-transfer barrier-crossing rates.

Experimental work probing the solvent dynamics aspects of these theories has involved (i) measurements of solvent relaxation times, such as with time-dependent fluorescence Stokes shifts,<sup>1,5</sup> and measurements of the solvent dependence of barrier-crossing rates based (ii) on excited-state initiation of electron-transfer reactions<sup>1,6</sup> and (iii) on ground-state heterogeneous (electrochemical) and homogeneous electron self-exchange processes.<sup>7-10</sup> To our knowledge, there have been no previous investigations of solvent dynamics effects in electron-transfer cross-reactions, in electron transfers at electron-transfer-active polymer surfaces, or in polymeric solvents. We report such studies here.

The solvent effect measurements are based on an electrontransfer cross-reaction at a redox polymer/solution interface:

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$$\frac{Pt/poly([Os]^{3+}) + [Fe(Me_2phen)_3]^{2+} \xrightarrow{\kappa_{12}}}{Pt/poly([Os]^{2+}) + [Fe(Me_2phen)_3]^{3+} (1)}$$

where poly([Os]<sup>3+</sup>) ( $E^{\circ}_{Os(111/11)} = 0.72$  V vs SSCE) is a thin-film polymer electroreductively deposited on a 10- $\mu$ m (diameter) Pt microelectrode, from a solution of the monomer [Os(bpy)<sub>2</sub>(vpy)<sub>2</sub>]<sup>2+</sup> (bpy = 2,2'-bipyridine, vpy = 4-vinylpyridine), and contacted with a [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> solution (Me<sub>2</sub>phen = 4,7-dimethyl-phenanthroline;  $E^{\circ}_{Fe(III/II)} = 0.89$  V vs SSCE). When  $E_{Pt}$  is made sufficiently positive of  $E^{\circ}_{Os(III/II)}$ , the limiting current flow measures the turnover of poly([Os]<sup>3+</sup>) states and thus the rate of reaction 1. From previous studies<sup>11</sup> of reaction 1 and related reactions in acetonitrile solvent, we know that (a) their rates vary with the reaction free energy  $E^{\circ}_{Fe(111/11)} - E^{\circ}_{Os(111/11)} = \Delta E^{\circ}$  in accord with classical theory<sup>2,3</sup> and (b) the redox polymer film is essentially impermeable to the bulky cationic metal complex, so that the energetically simple electron-transfer cross-reaction involves only the outermost layer of poly  $([Os]^{3+})$  sites on the redox polymer film surface.

In this study,  $k_{12}$  is measured by using seven monomeric solvents for the  $[Fe(Me_2phen)_3]^{2+}$  complex, to establish that reaction 1 responds to the solvent longitudinal relaxation times  $(\tau_{\rm L})$  in the manner expected.<sup>4</sup>  $k_{12}$  is additionally measured in a low molecular weight poly(ether) solvent, CH<sub>3</sub>O(CH<sub>2</sub>CH<sub>2</sub>O)<sub>8</sub>CH<sub>3</sub>, under the premise that relaxations of solvent dipoles that are attached to polymeric chains should be much slower than those characteristic of monomeric solvents and accordingly should provoke smaller values of  $k_{12}$ . Such a diminution is observed, providing insight into the dynamics of the polymer chain segment dipoles involved in the electron-transfer reaction.

We should note that reaction 1 is the rate-determining step in our scheme because it occurs in the thermodynamically disfavored direction. Proof that other steps-electron diffusion of poly-([Os]<sup>2+</sup>) states across the redox polymer film or the normally rapid back-reaction of reaction 1-do not interfere is given in the Appendix.

#### **Experimental Section**

Chemicals. Acetonitrile, methylene chloride, propylene carbonate (Burdick & Jackson, spectrochemical grade), and dimethoxyethane (Aldrich, anhydrous) were used as received. Acetone, pyridine (Aldrich), and dimethyl sulfoxide (Baker) were purified by standard methods.<sup>12</sup> Poly(ethylene glycol) dimethyl ether (Me<sub>2</sub>EG-400; Polyscience; nominal

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Table I. Solvent Properties and Electron-Transfer Kinetic Data for Cross-Reaction 1

solvent	€s <sup>a</sup>	€op <sup>a</sup>	$1/\epsilon_{op} - 1/\epsilon_s$	η,ª cp	$\tau_{\rm L}$ , b ps	$D_{\rm s}$ , c 10 <sup>6</sup> cm <sup>2</sup> /s	$\Delta E^{0,d}$ V	$k_{12}\Gamma, c_{m/s}$	k <sub>12</sub> , <sup>f</sup> M <sup>-1</sup> s <sup>-1</sup>	$\frac{k_{12}^{\text{corr},g}}{M^{-1} \text{ s}^{-1}}$
CH <sub>1</sub> CN	35.94	1.800	0.528	0.341	0.2	8.3	0.170	$1.7 \times 10^{-2}$	$1.7 \times 10^{5}$	$1.7 \times 10^{5}$
acetone	20.56	1.839	0.492	0.3029	0.3	8.7	0.200	$7.2 \times 10^{-3}$	$7.2 \times 10^{4}$	$1.3 \times 10^{5}$
CH <sub>2</sub> Cl <sub>2</sub>	8.93	2.020	0.383	0.4414	0.4	6.1	0.280	$1.9 \times 10^{-3}$	$1.9 \times 10^{4}$	$1.6 \times 10^{5}$
ру	12.91	2,272	0.363	0.884	1.2	2.5	0.200	$3.4 \times 10^{-3}$	$3.4 \times 10^{4}$	$6.1 \times 10^{4}$
DME	7.20	1.899	0.388	0.455	1.9	5.2	0.185	$2.7 \times 10^{-3}$	$2.7 \times 10^{4}$	$3.6 \times 10^{4}$
DMSO	46.45	2.183	0.437	1.991	2.35	1.5	0.170	$1.7 \times 10^{-3}$	$1.7 \times 10^{4}$	$1.7 \times 10^{4}$
PC	64.92	2.016	0.481	2.53	2.7	1.3	0.145	$1.3 \times 10^{-3}$	$1.3 \times 10^{4}$	$8.0 \times 10^{3}$
Me <sub>2</sub> PEG-400	9.16*	2.128	0.378	14.4	22	0.27	0.220	$2.4 \times 10^{-4}$	$2.4 \times 10^{3}$	$6.0 \times 10^{3}$

<sup>a</sup> From Organic Solvents, 4th ed.; Wiley-Interscience: New York, 1986. <sup>b</sup>Longitudinal relaxation time, from ref 7a,c and 18. <sup>c</sup>Diffusion coefficient of  $[Fe(Me_2phen)_3]^{2+}$ . <sup>d</sup> Formal potential difference,  $\Delta E^0 = E^0 F_{e(11/11)} - E^0 F_{Os(111/11)}$ . <sup>e</sup>Effective heterogeneous rate constant, uncorrected for variation in  $\Delta E^0$ . <sup>f</sup>Calculated with  $\Gamma = 1 \times 10^{-10}$  mol/cm<sup>2</sup> (monolayer) from  $k_{12}\Gamma$ . <sup>d</sup> Effective-transfer rate constant of reaction 1, normalized to  $\Delta E^0 = 170$  mV. <sup>h</sup> For  $\epsilon_0$ , assume the value for tetraglyme (*Industrial Solvent Handbook*; Noyes Data Corporation: Park Ridge, NJ, 1985);  $\epsilon_{op}$  was obtained from our refractive index measurement ( $n_D = 1.4587$  at 23 °C).

MW 400, av) was dried in vacuo (50 °C).  $LiClO_4$ ,  $Et_4NClO_4$ , and  $Bu_4NClO_4$  electrolytes were recrystallized and dried in vacuo (50 °C) and used throughout at 0.1 M concentrations.

Measurements. A 10- $\mu$ m (diameter) Pt working microelectrode and 26-gauge wire Pt auxiliary and Ag pseudoreference electrodes were sealed in a cylinder of insulating materials and polished so that the wire tips define the electrodes.<sup>13</sup> The 10- $\mu$ m Pt working electrode was coated with a thin film of poly([Os(by)<sub>2</sub>(vpy)<sub>2</sub>](PF<sub>6</sub>)<sub>2</sub>) by electropolymerization from 0.3-0.5 mM solutions of its monomer, cycling the potential through the ligand reduction waves rapidly (50 V/s) to minimize deposition beyond the microelectrode edge.<sup>14</sup> The quantity of deposited poly[Os] polymer,  $\Gamma_{T}$  (mol/cm<sup>2</sup>), was  $1 \times 10^{-8}$  mol/cm<sup>2</sup> or less, as determined from the charge under the Os(II/III) cyclic voltammetric wave in 0.1 M Et<sub>4</sub>NCIO<sub>4</sub>/CH<sub>3</sub>CN. All kinetic measurements were conducted in solvents containing 0.1 M electrolyte, each with a freshly prepared redox film and [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> solution. Electrochemical equipment was of standard design.<sup>15</sup> The viscosities of Me<sub>2</sub>PEG-400 and its mixtures with CH<sub>3</sub>CN, all containing 0.1 M LiCIO<sub>4</sub>, were measured by using Cannon Ubbelohde type viscometers in a temperature bath. All experiments were at 23 ± 1 °C.

#### Results

**Reaction 1. Electron-Transfer Kinetics.** Figure 1 presents the essential elements of the  $k_{12}$  measurement in 0.1 M Et<sub>4</sub>NClO<sub>4</sub>/CH<sub>3</sub>CN. Curve A is a cyclic voltammogram for the poly([Os]<sup>3+/2+</sup>) couple in the absence of [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup>, from which  $\Gamma_T$  is measured, and curves B and C show the oxidation of [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> at Pt microelectrodes coated<sup>16</sup> and not coated with a poly[Os] film. The potential of curve B is slightly more negative than that of curve C, which is more positive than that of curve A, both since the mediation reaction 1 is thermodynamically disfavored.<sup>17</sup> The curve B limiting current is jointly limited by the rate of reaction 1 and [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> diffusion to the redox polymer/solution interface,<sup>17</sup> whereas the curve C current is limited only by the latter process and is larger. The curve B limiting current is related to the characteristic currents  $i_{kin}$  and  $i_{MT}$  for reaction 1 and [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> diffusion, respectively, by<sup>17</sup>

$$(i_{\rm lim})^{-1} = (i_{\rm MT})^{-1} + (i_{\rm kin})^{-1} = (4rnFD_{\rm s}C_{\rm s})^{-1} + (nF\pi r^2 k_{12}\Gamma C_{\rm s})^{-1}$$
(2)

where  $C_s$  and  $D_s$  are the solution concentration and diffusion coefficient of  $[Fe(Me_2phen)_3]^{2+}$ , r is the microelectrode radius,  $k_{12}$  is the second-order reaction 1 rate constant (cm<sup>3</sup>/(mol s)), and  $\Gamma$  is the quantity of osmium complex in the outermost layer



Figure 1. Microelectrode voltammetry of (a) 10- $\mu$ m Pt disk coated with poly[Os],  $\Gamma_T = 5.7 \times 10^{-9}$  mol/cm<sup>2</sup>, in 0.1 M Et<sub>4</sub>NClO<sub>4</sub>/CH<sub>3</sub>CN, 50 mV/s; (b) the same electrode in 0.22 mM [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup>, 0.1 M Et<sub>4</sub>NClO<sub>4</sub>/CH<sub>3</sub>CN, 5 mV/s; (c) naked 10- $\mu$ m Pt disk in 0.22 mM [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup>, 0.1 M Et<sub>4</sub>ClO<sub>4</sub>/CH<sub>3</sub>CN, 5 mV/s; (d) 10- $\mu$ m Pt disk coated with poly[Os],  $\Gamma_T = 4.9 \times 10^{-9}$  mol/cm<sup>2</sup>, 0.1 M LiClO<sub>4</sub>/Me<sub>2</sub>PEG-400, 10 mV/s; (e) the same electrode in 1.7 mM [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup>, 0.1 M LiClO<sub>4</sub>/Me<sub>2</sub>PEG-400, 5 mV/s; (f) naked 10- $\mu$ m Pt disk in 1.7 mM [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup>, 0.1 M LiClO<sub>4</sub>/Me<sub>2</sub>PEG-400, 5 mV/s; (f) naked 10- $\mu$ m Pt disk in 1.7 mM [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup>, 0.1 M LiClO<sub>4</sub>/Me<sub>2</sub>PEG-400, 5 mV/s. Cross represents 0 A and 0 V vs SSCE using Os(III/II) peak as an internal reference (in CH<sub>3</sub>CN,  $E^{\circ}_{Os(III/II)} = 0.72$  V vs SSCE).

of the poly[Os] film. The product  $k_{12}\Gamma$  represents the heterogeneous electron-transfer rate constant (cm/s) for [Fe-(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> oxidation at the redox polymer surface. The mass transport current  $i_{MT}$  is measured from curve C.

Figure 1B and eq 2 yield  $k_{12}\Gamma = 0.017$  cm/s for reaction 1 in acetonitrile solvent, in good agreement with a previous rotated disk electrode measurement<sup>11</sup> in the same solvent  $(k_{12}\Gamma = 0.019$ cm/s). Analogous experiments in pyridine, dimethoxyethane, methylene chloride, acetone, dimethyl sulfoxide, and propylene carbonate solutions of  $[Fe(Me_2phen)_3]^{2+}$  reveal completely parallel behavior except that the value of  $k_{12}\Gamma$  varies with the solvent as reported in Table I along with some relevant solvent properties and  $D_s$  values. Table I also shows  $k_{12}$  values calculated by taking  $\Gamma$  as a monolayer,  $1 \times 10^{-10}$  mol/cm<sup>2</sup>, and  $k_{12}^{corr}$  values after normalizing  $k_{12}$  to a common reaction free energy (correcting for  $\Delta E^{\circ}$  variation from solvent to solvent, Table I). Values of  $k_{12}\Gamma$ for reaction 1 do not vary with the redox polymer coverage  $\Gamma_T$ (from 1 to  $8 \times 10^{-9}$  mol/cm<sup>2</sup>) in different solvents, as expected if reaction 1 involves only the outermost monolayer of poly[Os] sites and on the basis of our previous analysis<sup>11</sup> of this aspect of the interfacial reaction.

Me<sub>2</sub>PEG-400 is a viscous ( $\eta = 14.4$  cp for 0.1 M LiClO<sub>4</sub> solution) polymer melt that readily dissolves LiClO<sub>4</sub> electrolyte and many electroactive monomers. It has a wide potential window and generally low background currents.<sup>19</sup> Using Me<sub>2</sub>PEG-400

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<sup>(19)</sup> Longmire, M. L.; Wooster, T. T.; Watanabe, M.; Zhang, H.; Murray, R. W. Manuscript in preparation.



Figure 2. Panel A:  $[Fe(Me_2phen)_3]^{2+}$  diffusion coefficient,  $D_s$ , vs reciprocal viscosity, eqn. 3, of the monomer solvents. Panel B:  $k_{12}\Gamma$  of reaction 1 vs reciprocal monomer solvent viscosity.

polymer as solvent produces voltammetric behavior (Figure 1D-F) like that in the monomeric solvents, except that the limiting currents for [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> oxidation are much smaller on both poly[Os]-coated (curve E) and naked (curve F) electrodes. Analysis with eq 2 shows (Table I) that both the  $[Fe(Me_2phen)_3]^{2+}$ diffusion coefficient  $D_s$  and the reaction 1  $k_{12}\Gamma$  and  $k_{12}^{corr}$  values are smaller in the polymer solvent. It is noteworthy that  $k_{12}\Gamma$  and  $k_{12}^{corr}$  are much smaller in Me<sub>2</sub>PEG-400 than in its close chemical relative, DME, which is an explicit illustration of slowed electron-transfer dynamics in a polymeric solvent.

 $k_{12}\Gamma$  rate constants were reproducible typically to 30% and 50% in the monomer and polymer solvents, respectively.

Diffusion Coefficients and Viscosity. In the monomer solvents, diffusion coefficients for [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> vary (Figure 2A) with viscosity according to the Stokes-Einstein equation

$$D = kT/(6\pi R_{\rm h}\eta) \tag{3}$$

where  $\eta$  is viscosity, k the Boltzmann constant, and  $R_{\rm h}$  the hydrodynamic radius of the complex. Figure 2A gives  $R_h = 8.2$  Å for  $[Fe(Me_2phen)_3]^{2+}$ ; this value is needed in the theoretical analysis given below.

Systematic variation in heterogeneous and homogeneous electron-transfer rate constants with solution viscosity has been reported on several occasions,  $^{8b-c,9c,10}$  Figure 2B shows that no such regular viscosity dependence is found for  $k_{12}\Gamma$  in reaction 1. Clearly other factors are involved.

Comparison to Theory for Monomer Solvents. An outer sphere homogeneous electron self-exchange rate constant is classically given by2,3

$$k = K_{\text{pre}\kappa_{\text{el}}\nu_{\text{n}}} \exp[-\Delta G_{\text{Os}}^{*}/RT] = K_{\text{pre}\kappa_{\text{el}}\nu_{\text{n}}} \exp[-(Ne^{2}/16\pi\epsilon_{0})(1/a - 1/R_{\text{h}})(1/\epsilon_{\text{opl}} - 1/\epsilon_{\text{s}})/RT]$$
(4)

where  $\Delta G_{Os}^{*}$  is activation free energy determined by the outer shell or solvent reorganization energy, a is the equivalent reactant radius,  $R_h$  is the reactant internuclear separation (we assume  $R_h$ = 2a),  $\epsilon_{opt}$  and  $\epsilon_s$  are the optical and static dielectric constants for the surrounding solvent, respectively,  $K_{pre}$  is the equilibrium constant for forming the precursor (reactant pair) complex prior to electron transfer,<sup>20</sup>  $\kappa_{el}$  is the electronic transmission coefficient,<sup>21</sup>

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Figure 3. Plots according to eqs 4 and 5 for  $k_{12}$  (panel A) and  $k_{12}^{corr}$ (panel B). Theoretical line (-) has slope based on  $R_h = a_1 = 8.2$  Å for  $[Fe(Me_2phen)_3]^{2+}$  and  $a_2 = 7$  Å for<sup>11</sup> poly( $[Os]^{3+}$ ) sites and is forced through the acetonitrile point, for comparison to experimental (---).

and  $v_n$  describes the net dynamics along the nuclear reaction coordinate in the vicinity of the barrier top.

The classical treatment mainly conveys the solvent influence on electron transfer by the dielectric constant term in  $\Delta G_{Os}^*$ . However, it is obvious by inspection that the Table I data contain no systematic variation of  $(1/\epsilon_{opt} - 1/\epsilon_s)$  with  $k_{12}$  and  $k_{12}^{corr}$ . That is, while we know<sup>11</sup> that the classical theory satisfactorily describes the free energy dependence of reaction 1, it does not account for the effect of different solvents on its rate.

Recent treatments<sup>4.7</sup> have emphasized that the dynamics of solvent reorganization provide an important contribution to  $v_n$ . At least for a Debye fluid (that having a single relaxation time,  $\tau_{\rm D}$ ), dielectric continuum treatment of the barrier-crossing dynamics in "over-damped" solvents yields the relation

$$\nu_{\rm n} = \tau_{\rm L}^{-1} (\Delta G_{\rm Os}^{*} / RT)^{1/2}$$
 (5)

where  $\tau_{\rm L}$  is the solvent longitudinal dielectric relaxation time, related to the Debye relaxation time  $\tau_D$  by<sup>22</sup>

$$\tau_{\rm L} = \tau_{\rm D}(\epsilon_{\rm \infty}/\epsilon_{\rm s}) = (\epsilon_{\rm \infty}/\epsilon_{\rm s})4\pi a_{\rm s}^{3}\eta/kT \tag{6}$$

where  $\epsilon_{\infty}$  is the high-frequency dielectric constant and  $a_{\rm s}$  the radius of the solvent molecule. These relations introduce a solvent dielectric constant and relaxation time dependence in the preexponential of eq 4.

The rate  $(k_{12})$  of an electron-transfer cross-reaction (i.e., reaction 1) is<sup>2,3</sup> related to the self-exchange rate constants  $k_{11}$  and  $k_{22}$  of the reactant couples and the reaction free energy (e.g., the equilibrium constant  $K_{12}$ ) by

$$k_{12} = (k_{11}k_{22}K_{12}f)^{1/2} \qquad \log f = (\log K_{12})^2 / [4 \log (k_{11}k_{22}/Z^2)]$$
(7)

where Z is the electron-transfer preexponential factor having dimensions of collision frequency, and in the present case,  $f \approx 1.2^3$ 

From these relations and following analogous<sup>9a-b,24</sup> treatments of heterogeneous and homogeneous electron transfers, one predicts that, at constant  $\Delta E^{\circ}$  (i.e., constant  $K_{12}$ ), log  $[k_{12}\tau_{\rm L}/(\epsilon_{\rm op}^{-1}-\epsilon_{\rm s}^{-1})^{1/2}]$ 

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<sup>(21)</sup> When solvent dynamics become important, κ<sub>el</sub> tends to unity (i.e., adiabatic or near adiabatic conditions.) See: McManis, G. E.; Nielson, R. M.; Gochev, A.; Weaver, M. J. J. Am. Chem. Soc. **1989**, 111, 5533.
(22) Smyth, C. P. Dielectric Behavior and Structure, McGraw-Hill: New Net Proceedings.

York, 1955.

<sup>(23)</sup> The assumption that f = 1 introduces <10% error in  $k_{12}$ ; see eq 4, ref 11.

should vary linearly with  $(\epsilon_{op}^{-1} - \epsilon_s^{-1})$  with slope  $(2.303Ne^2/32\pi\epsilon_0 RT)(a_1^{-1} + a_2^{-1})$ . This analysis is applied to the experimental  $k_{12}$  data (Table I) in Figure 3A, where the solid line is forced, with a theoretical slope, through the point for acetonitrile. The experimental-theory comparison is poor; the points are scattered, and the theoretical (--) and experimental (---) slopes differ. However, this comparison neglects the variation in reaction free energy with solvent seen in the  $\Delta E^{\circ}$  values in Table I. Because reaction 1 electron cross-transfers are known<sup>11</sup> to obey the classical free energy relationships, we can normalize the reaction with eq 7 to a rate  $k_{12}^{\text{corr}}$  at a common free energy  $\Delta E^{\circ} = 0.170 \text{ V}$ , the value in acetonitrile. The  $k_{12}^{corr}$  results (Table I) give (Figure 3B) a now impressive linear correlation between log  $[k_{12}^{\text{corr}}\tau_{L}/(\epsilon_{op}^{-1})]$  $-\epsilon_s^{-1}$  and  $(\epsilon_{op}^{-1} - \epsilon_s^{-1})$  and a near-theoretical slope (experimental slope -3.61, theoretical -4.01). This agreement rivals that found in previous<sup>7</sup> analysis of solvent effects on electron-transfer reaction rates. (Studies of ionic strength effects by Lewis and Obeng<sup>25</sup> indicate that even better agreement could be sought by extrapolating kinetic data to zero electrolyte concentration, but this was not attempted here.) The result in Figure 3B demonstrates that the solvent dipole reorganization dynamics around the poly([Os]<sup>3+</sup>) sites embedded in the redox polymer/monomer solvent interface are not especially different from those of monomeric solutes. This finding is consistent with the previous<sup>11</sup> observations on reaction 1 free energy effects.

**Comparison to Theory for Me<sub>2</sub>PEG-400 Polymer Solvent.** As noted above, the Figure 1D-F voltammetric measurements in M<sub>2</sub>PEG-400 solvent produce a reaction 1 rate constant  $k_{12}^{\text{corr}}$  that is smaller than observed in any of the monomer solvents (Table I). The Appendix will show that the requirements of eq 2 for measurement of  $k_{12}\Gamma$  are satisfactorily met in this medium, so the depressed cross-electron-transfer rate is a real solvent effect on the electron-transfer kinetics. We interpret the solvent effect as being of the same nature as that demonstrated in monomer solvents (i.e., Figure 3B) but in this case governed by motions of dipoles located on linear solvent chains that are sluggish as compared to those in monomeric fluids.

In its melt state, the poly(ether) Me<sub>2</sub>PEG-400 is expected to exist as a random coil. The displacements or reorientations of the ether dipoles that are relevant to the electron-transfer barrier crossing in reaction 1 undoubtedly do not involve movement of the entire random coil but instead rearrangements via main-chain bond rotations of one or more segments  $-(CH_2CH_2O)-$  or subsegments of the ensemble of coiled molecules comprising the solvation shell of the electron-transfer reactants. Additionally, there may be contributions from the electrolyte dipoles, either directly or more likely through their indirect, coordinative chain cross-linking influence<sup>26</sup> on chain segment mobility. The polymer solvent dynamics are thus considerably more complex than monomeric solvent relaxations.

Since relaxation time data equivalent to those for monomer solvents (Table I) are not known, we chose to employ Figure 3B as a working plot for reaction 1, inquiring what *effective* relaxation " $\tau_L$ " value for the polymer solvent would place the experimental  $k_{12}$  result in Me<sub>2</sub>PEG-400 solvent into agreement with the monomer solvent data. Approximating the static dielectric constant of Me<sub>2</sub>PEG-400 with that of tetraglyme (Table I) produces a " $\tau_L$ " value of 22 ps (Table I) for Me<sub>2</sub>PEG-400. This " $\tau_L$ " measures the dynamics for relaxation of ether dipoles located on the poly(ether) chains that are pertinent to the electron-transfer barrier-crossing event in MPEG-400 solvent.

While slower than the relaxation times (Table I) for monomer solvents, the 22-ps " $\tau_L$ " for Me<sub>2</sub>PEG-400 nonetheless represents a rather fast time scale for polymeric materials. For example, dielectric relaxation loss maxima for poly(propylene oxide), PPO, occur in a much slower (ns) regime<sup>27</sup> and are thought to reflect



Figure 4.  $[Fe(Me_2phen)_3]^{2+}$  diffusion coefficient,  $D_s$ , vs reciprocal viscosity,  $1/\eta$ , of 0.1 M LiClO<sub>4</sub>/CH<sub>3</sub>CN/Me<sub>2</sub>PEG-400 mixtures ( $\Delta$ );  $k_{12}^{00TT}$  vs reciprocal viscosity (O) and mixture mole fraction (moles of CH<sub>3</sub>CN vs moles of EO unit) ( $\odot$ ).

correlated reorientations of ensembles of dipoles. NMR correlation times for polymer chain motion in poly(ethylene oxide)/NaClO<sub>4</sub> are<sup>28</sup> ~400 ps at 69 °C. Polymer dynamics measurements that sense the more rapid structural dipolar relaxations in poly(ethers) appear to be Brillouin scattering data, for which structural relaxation time ( $\tau_s$ ) data has been gathered for poly(propylene oxide) chains.<sup>26a,27,29,30</sup>  $\tau_s$  values in pure PPO-4000 ( $\tau_s \approx 50$  ps) and in solutions containing NaSO<sub>3</sub>CF<sub>3</sub> ( $\tau_s \approx 100$  ps; Na:O = 1:16) all lie substantially above the 10<sup>-12</sup>-s time range typical of monomer fluid  $\tau_L$  values (Table I). This is of course a very approximate comparison, but the reaction 1  $\tau_L$  time scale is nonetheless encouragingly near known time scales<sup>26a,27,29,30</sup> for local structural motions that lead to dipolar changes.

It is worth emphasizing that the " $\tau_{\rm L}$ " derived for Me<sub>2</sub>PEG-400 in reaction 1 is a *microscopic* quantity and bears no correspondence to a time constant estimated by using the *macroscopic* viscosity of the polymer solvent and eq 6. For example, taking a radius of 5 Å (given by root mean square end-to-end distance of polymer chains) for the coiled polymer, we predict via eq 6 a  $\tau_{\rm L}$  of ca. 2  $\times 10^4$  ps, which is far larger than that observed.

Mixtures of Me<sub>2</sub>PEG-400 and Acetonitrile. Mixtures of acetonitrile monomer solvent and Me<sub>2</sub>PEG-400 polymer solvent were studied to further explore the effects of polymeric media. Acetonitrile is a relatively "fast" solvent with a longitudinal relaxation time ( $\tau_L = 0.2$  ps) 100-fold smaller than that for Me<sub>2</sub>PEG-400. Of interest was whether a small population of "fast dipoles" in a mixture would negate the slowing of the electron-transfer rate attendant to the sluggish poly(ether) dipoles.

Results in the solvent mixtures are presented in Figure 4. The diffusion coefficient,  $D_s$ , of  $[Fe(Me_2phen)_3]^{2+}$  continues  $(\Delta)$  to follow, roughly, solution viscosity according to the macroscopic viscosity law (eq 3). Figure 4 (O) also compares  $k_{12}\Gamma$  to reciprocal viscosity; the result is similar to observations by Bard et al.<sup>10</sup> except that the rate constant climbs sharply on the low-viscosity side. Most significantly, the rate constant  $k_{12}\Gamma$  of reaction 1 (corrected for change in  $\Delta E^\circ$ ) is responsive ( $\bullet$ ) to rather small amounts of Me<sub>2</sub>PEG-400 in CH<sub>3</sub>CN, decreasing sharply and then more gradually as pure Me<sub>2</sub>PEG-400 is approached.

Figure 4 ( $\bullet$ ) clearly shows that dipolar relaxation times for CH<sub>3</sub>CN and Me<sub>2</sub>PEG-400 do not average together on a simple mole fraction basis, nor does the "fast" CH<sub>3</sub>CN dominate the electron-transfer behavior. Inferences possibly drawn from these results are (a) that somehow a relatively small concentration of Me<sub>2</sub>PEG-400 chains greatly slows the relaxation dynamics of

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Figure 5. Scheme for reaction 1 and concentration-distance diffusion profiles for poly[Os] sites and metal complex.



Figure 6. Microelectrode voltammograms (10 mV/s) of 10.7 mM decamethylferrocene, Cp<sub>2</sub>\*Fe, in 0.1 M LiClO<sub>4</sub>/Me<sub>2</sub>PEG-400 at (A) 10- $\mu$ m Pt disk coated with poly[Os] film,  $\Gamma_T = 1.3 \times 10^{-9}$  mol/cm<sup>2</sup>, and (B) not coated, and the ferrocene concentration dependence of the limiting currents.

CH<sub>3</sub>CN, (b) the redox polymer surface and the  $[Fe(Me_2phen)_3]^{2+}$  reactants are preferentially solvated by coils of Me<sub>2</sub>PEG-400 so that the local reaction environment becomes relatively polymerlike, and (c) the rate of dipole relaxations on poly(ether) chains is greatly enhanced in mixtures with CH<sub>3</sub>CN. The first of these inferences does not seem very plausible in comparison to the latter two. Indeed, Brillouin scattering results exist that show a relaxation rate increase of poly(propylene oxide) (PPO) upon dilution with methyl cyclohexane.<sup>27</sup> We are unable to carry the analysis further at this point, however.<sup>31</sup> Acknowledgment. This research was supported in part by grants from the Department of Energy and the National Science Foundation.

### Appendix

Justification of Equation 2 for Redox Polymer/Polymer Solvent Interface. We establish here the validity of eq 2 for reaction 1 in a polymer solvent. A crucial issue is whether electron transport in the redox polymer film  $(D_e)$  and metal complex diffusion in the polymer solvent  $(D_s)$  are sufficiently rapid that poly $([Os]^{2+})$ and  $[Fe(Me_2phen)_3]^{3+}$  exit the redox polymer/polymer solvent interface rather than (thermodynamically favored) back-react. The prior evidence is ample<sup>11</sup> that the complex does not penetrate the redox polymer film. Figure 5 shows the concentration-distance diffusion profiles for metal complex and poly[Os] sites set up by the cross-exchange reaction. When  $E_{Pi}$  is much more positive than  $E^o_{Os(111/11)}$ 

$$i_{\lim}/nFA = (D_{e}/d)C_{Os(11)(x=d)} = (D_{e}/d)[C_{T} - C_{Os(111)(x=d)}]$$
(8)
$$= (D_{e}/\delta)C_{Ee(111)(x=d)} = (D_{e}/\delta)[C_{e} - C_{Ee(11)(x=d)}]$$
(9)

$$= k_{12}C_{Os(111)(x=d)}C_{Fe(11)(x=d)} - k_{21}C_{Os(11)(x=d)}C_{Fe(111)(x=d)}$$
(10)

where  $C_s$  and  $C_T$  are the initial concentrations of  $[Fe(Me_2phen)_3]^{2+}$ and poly[Os],  $k_{12}/k_{21} = K_{eq} \ll 1$ , and d and  $\delta$  are the thicknesses of the redox polymer film and diffusion layer in the solution. Defining characteristic currents  $i_{MT} = nFAD_sC_s/\delta$ ,  $i_{ET} = nFA D_eC_T/d$ , and  $i_{kin} = nFAk_{12}\Gamma C_s$  gives<sup>17b,c</sup>

$$1 = i_{\rm lim}/i_{\rm kin} + i_{\rm lim}/i_{\rm MT} + i_{\rm lim}/i_{\rm ET} + (i_{\rm lim}/i_{\rm ET})(i_{\rm lim}/i_{\rm MT})[(1/K_{\rm EQ}) - 1]$$

Equation 2 corresponds to the case where the two right-hand terms are negligible. In the present case, in MPEG-400,  $D_e = 10^{-8}$  cm<sup>2</sup>/s<sup>11</sup>,  $D_s = 3 \times 10^{-7}$  cm<sup>2</sup>/s,  $d \approx 10^{-6}$  cm,  $i_{lim} = 10^{-11}$  A,  $i_{MT} = 5 \times 10^{-11}$  A, and  $i_{ET} \approx 8 \times 10^{-7}$  A, so that  $i_{lim}/i_{ET} = 1.2 \times 10^{-5}$ ,  $i_{lim}/i_{MT} = 0.2$ , and the fourth right-hand term is  $2 \times 10^{-3}$ , demonstrating that the back-reaction of reaction 1 detracts from the measured rate to a negligible extent.

Since the prior measurement of  $D_e$  in poly[Os] was not in MPEG-400 solvent, we explicitly show that  $i_{\rm ET} \gg i_{\rm lim}$  by using a substrate with a formal potential *less* positive than that of Os(II/III), making its electron-transfer cross-reaction now fast compared to the transport steps. Figure 6 shows voltammograms and their concentration dependence for decamethylferrocene, Cp<sub>2</sub>\*Fe ( $E^{\circ \prime} = 0.13$  V vs Ag wire in Me<sub>2</sub>PEG-400/LiClO<sub>4</sub>), at naked and at poly[Os]-coated electrodes. The currents at the naked electrode,  $i_{\rm MT}$ , and those at the coated electrode,  $i_{\rm lim}$ , clearly the same, vary linearly with Cp<sub>2</sub>\*Fe concentration. Were electron diffusion in the redox polymer film current-controlling, the  $i_{\rm lim}$ values would become *independent* of [Cp<sub>2</sub>\*Fe]. Since the largest  $i_{\rm lim}$  values in Figure 6 are more than 10<sup>2</sup> times larger than  $i_{\rm lim}$ values measured for [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> in Figure 1E,  $i_{\rm ET}$  limitations there can be unambiguously ruled out.

<sup>(31) (</sup>a) We observe, from refractive index measurements of the solvent mixtures, a nonlinear increase in  $1/\epsilon_{op}$  with mole percent CH<sub>3</sub>CN that parallels (but is smaller than) that in Figure 4. We also attempted to examine differential solvation, following Hupp<sup>31b</sup> and Meyer,<sup>31c</sup> by observing the MLCT transition in the [Fe(Me<sub>2</sub>phen)<sub>3</sub>]<sup>2+</sup> complex, but the small change between CH<sub>3</sub>CN and Me<sub>2</sub>PEG-400 (only 5 nm, blue shift in the latter) prevented any clean inferences. (b) Blackbourn, R. L.; Hupp, J. T. J. Phys. Chem. 1988, 92, 2817. (c) Kober, E. M.; Sullivan, B. P.; Meyer, T. J. Inorg. Chem. 1984, 23, 2098.